

# Canted ferrimagnetism and giant coercivity in the nonstoichiometric double perovskite $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$

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The nonstoichiometric double perovskite oxide  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  was synthesized by solid-state reaction and its crystal and magnetic structures were investigated by powder x-ray and neutron diffraction.  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  crystallizes in the monoclinic double perovskite structure (general formula  $A_2B'B'O_6$ ) with space group  $P2_1/n$ , where the  $B$  site is fully occupied by Ni and the  $B'$  site by 19% Ni and 81% Os atoms. Using x-ray absorption spectroscopy an  $\text{Os}^{4.5+}$  oxidation state was established, suggesting the presence of about 50% paramagnetic  $\text{Os}^{5+}$  ( $5d^3$ ,  $S = 3/2$ ) and 50% nonmagnetic  $\text{Os}^{4+}$  ( $5d^4$ ,  $J_{\text{eff}} = 0$ ) ions at the  $B'$  sites. Magnetization and neutron diffraction measurements on  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  provide evidence for a ferrimagnetic transition at 125 K. The analysis of the neutron data suggests a canted ferrimagnetic spin structure with collinear  $\text{Ni}^{2+}$ -spin chains extending along the  $c$  axis but a noncollinear spin alignment within the  $ab$  plane. The magnetization curve of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  features a hysteresis with a very high coercive field,  $H_C = 41$  kOe, at  $T = 5$  K, which is explained in terms of large magnetocrystalline anisotropy due to the presence of Os ions together with atomic disorder. Our results are encouraging to search for rare-earth-free hard magnets in the class of double perovskite oxides.

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## I. INTRODUCTION

Double perovskite oxides  $A_2BB'O_6$  containing  $3d$ - $5d(4d)$  transition-metal ions at the  $B$  and  $B'$  sites are attracting great attention, due to their interesting physical properties. High-temperature ferrimagnetic half-metals and ferro- or ferrimagnetic insulators, as well as materials with large magnetoresistance and exchange bias, are found in this class of compounds [1–9]. They thus show prospects for spintronic applications. Most of these compounds contain  $5d$  elements with  $5d^2$  and  $5d^3$  electronic configurations. Here, the interesting properties arise from competing interactions within and between the  $3d$  and  $5d$  sublattices. Studies on  $5d^4$  systems are rare, probably because of the expected nonmagnetic ground state for  $5d^4$  ions. In the presence of strong spin-orbit coupling (SOC), the orbital angular momentum associated with the three  $t_{2g}$  orbitals in an octahedral crystal field ( $l = 1$ ) entangles with the spin moments of the electrons which results in an upper  $j_{\text{eff}} = 1/2$  doublet and a lower  $j_{\text{eff}} = 3/2$  quadruplet. The formation of a SOC-assisted Mott insulating state in  $\text{Sr}_2\text{IrO}_4$  ( $\text{Ir}^{4+}$ ,  $t_{2g}^5$ ) is understood within this picture [10]. According to this scenario, the ground state of a  $5d^4$  system in an octahedral crystal field is a trivial singlet with four electrons filling the lower quadruplet, resulting in a nonmagnetic state with total angular momentum  $J_{\text{eff}} = 0$ . Many studies focus on  $\text{Ir}^{5+}$  compounds, such as  $A_2B\text{IrO}_6$  ( $A = \text{Sr}, \text{Ba}$ ;  $B = \text{Sc}, \text{Y}$ ) [11–17]. In contrast to the expectations for  $J_{\text{eff}} = 0$ , long-range magnetic order at 1.3 K was reported for  $\text{Sr}_2\text{YIrO}_6$  [11,12] and considered as

evidence for intermediate-strength rather than strong spin-orbit coupling which is usually anticipated for  $5d$  ions. These results, however, have been challenged and the observed magnetism in such materials may arise from defects and/or nonstoichiometry [13,18].

Besides  $\text{Ir}^{5+}$ ,  $\text{Os}^{4+}$  ions also adopt the  $5d^4$  electronic configuration. The singlet ground-state magnetism was discussed in  $\text{AOsO}_3$  ( $A = \text{Ca}, \text{Sr}, \text{Ba}$ ) [19] and  $\text{R}_2\text{Os}_2\text{O}_7$  ( $R = \text{Y}, \text{Ho}$ ) [20]. In this work, targeting at an  $\text{Os}^{4+}$  double perovskite  $\text{La}_2\text{NiOsO}_6$ , we synthesized a nonstoichiometric phase  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  with  $\text{Ni}^{2+}$  ions at the  $B$  site and a mixture of  $\text{Ni}^{2+}$ ,  $\text{Os}^{4+}$ , and  $\text{Os}^{5+}$  ions at the  $B'$  site of the double perovskite structure. It was verified by x-ray absorption spectroscopy (XAS) that the valence state of Os is about  $4.5+$ . Magnetization and neutron diffraction studies revealed that  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  shows a ferrimagnetic transition at 125 K and features an extraordinary broad hysteresis with a giant coercivity of 41 kOe at 5 K. These results are encouraging for the search of hard magnetic materials in the class of  $3d/5d$  double perovskites.

## II. EXPERIMENT

A polycrystalline sample of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  was synthesized by solid-state reaction from  $\text{La}_2\text{O}_3$ ,  $\text{NiO}$ , and  $\text{Os}$ .  $\text{La}_2\text{O}_3$ ,  $\text{NiO}$ , and  $\text{Os}$  in a molar ratio 1 : 1 : 1 were well ground together and pressed into a pellet. This was loaded into a corundum crucible which was placed into a silica tube

along with a second corundum crucible containing  $\text{MnO}_2$  (Alfa 99.9%). The silica tube was then sealed under dynamic vacuum using a  $\text{H}_2/\text{O}_2$  torch, and heated at  $1250^\circ\text{C}$  for 48 h in a tube furnace.  $\text{MnO}_2$  decomposes into one-half  $\text{Mn}_2\text{O}_3$  + one-fourth  $\text{O}_2$  at  $550^\circ\text{C}$  and acts as an oxygen source for the reaction. The molar ratio of Os and  $\text{MnO}_2$  is 1 : 4. Small pieces of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  were cut from the synthesized pellet and finely ground to a fine powder, which was characterized by powder x-ray diffraction using a Huber G670 camera [Guinier technique,  $\lambda = 1.54056 \text{ \AA}$  ( $\text{Cu-K}\alpha_1$ )]. A powder pattern was collected in the  $2\theta$  range between  $10.9^\circ$  and  $100.2^\circ$ . A scanning electron microscope (SEM) (Philips XL30) with an attached energy dispersive x-ray spectrometer (EDX) was used for elemental analysis.

The Os- $L_3$  XAS spectra of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$ , an  $\text{Os}^{5+}$   $\text{Sr}_2\text{FeOsO}_6$  reference, and an  $\text{Os}^{4+}$  reference  $\text{La}_2\text{MgOsO}_6$  were measured in transmission geometry at the beamline BL07A at the National Synchrotron Radiation Research Center in Taiwan.

Neutron powder diffraction experiments of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  were carried out on the instruments E2, E6, and E9 at the BER II reactor of the Helmholtz-Zentrum Berlin. The E9 instrument uses a Ge monochromator selecting the neutron wavelength  $\lambda = 1.309 \text{ \AA}$ , while the E2 and E6 instruments use a pyrolytic graphite (PG) monochromator selecting the neutron wavelength  $\lambda = 2.38$  and  $2.417 \text{ \AA}$ , respectively. In order to investigate in detail the crystal structure of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  at 3.2 and 160 K, well below and above the magnetic ordering temperature, neutron powder patterns were recorded on the E9 instrument between the diffraction angles  $7.5^\circ$  and  $141.8^\circ$ . For a detailed analysis of the magnetic structure and its temperature dependence, we have collected powder patterns between 1.7 and 134 K on the E6 instrument between the diffraction angles  $5^\circ$  and  $136.4^\circ$ . In addition, powder patterns at 1.7 and 150 K with higher counting rate and better instrumental resolution in the  $2\theta$  range between  $11.8^\circ$  and  $87.3^\circ$  were obtained on the E2 instrument.

Rietveld refinements of the powder diffraction data were carried out with the program FULLPROF [21]. In the case of the data analysis of x-ray diffraction data, we used the atomic scattering form factors provided by the program. For the refinements of the neutron powder data the nuclear scattering lengths  $b(\text{O}) = 5.803 \text{ fm}$ ,  $b(\text{La}) = 8.24 \text{ fm}$ ,  $b(\text{Ni}) = 10.3 \text{ fm}$ , and  $b(\text{Os}) = 10.7 \text{ fm}$  were used [22]. The magnetic form factors of the Fe and Os atoms were taken from Refs. [23,24].

Using a polycrystalline piece, the electrical resistivity ( $\rho$ ) was measured with direct current (0.1 mA) in a four-point in-line arrangement (PPMS, Quantum Design). Electrical contacts were made with Au wires and silver paste. The temperature dependence of the magnetic susceptibility was measured in a SQUID magnetometer (MPMS-5T, Quantum Design). The measurements were conducted in warming after zero-field cooling (ZFC) and during field cooling (FC) in the temperature range 2–300 K under applied magnetic fields of 10 kOe. The high-temperature magnetic susceptibility (300–565 K) was measured using the same SQUID magnetometer with the oven option. Isothermal magnetization curves were initially recorded for fields up to  $\pm 50$  kOe at temperatures of 5 and 150 K using the same SQUID magnetometer. As the magnetization curves turned out to be unsaturated at 5 K and

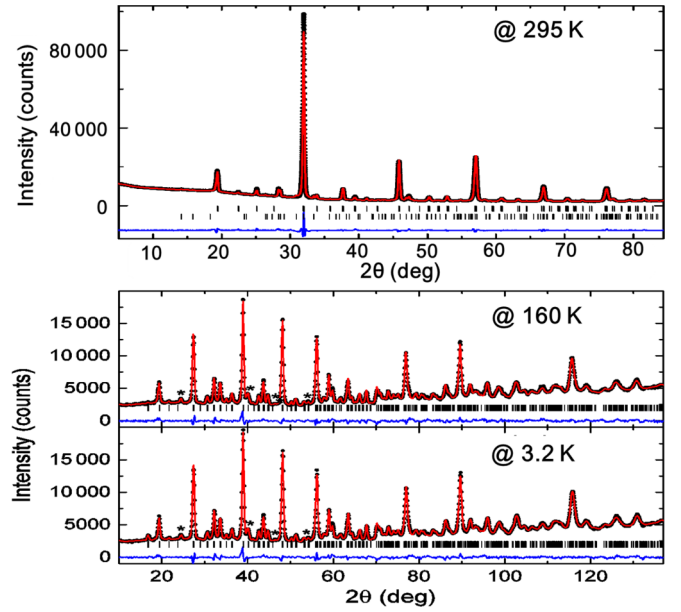


FIG. 1. Top: Rietveld refinement of the x-ray powder diffraction data of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  collected at 295 K. The vertical bars indicate the positions of the nuclear Bragg reflections for  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  and impurity phase  $\text{La}_3\text{OsO}_7$ . Bottom: Rietveld refinements of the powder neutron diffraction data of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  collected at 160 and 3.2 K (bottom). The strongest peaks of the impurity phase are marked with an asterisk.

showed a broad hysteresis,  $M(H)$  curves were further measured for fields up to  $\pm 140$  kOe at 5 K using a Physical Property Measurement System (PPMS, Quantum Design).

### III. RESULTS

#### A. Crystal structure

The room-temperature crystal structure of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  was investigated by x-ray powder diffraction as shown in Fig. 1 (top).  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  was successfully refined in the monoclinic space group  $P2_1/n$  (No. 14, standard setting  $P2_1/c$ ). In this space group, the La and the three O atoms (O1, O2, and O3) occupy the Wyckoff position  $4e(x,y,z)$ , while the Ni and Os atoms are at the positions  $2c(\frac{1}{2},0,\frac{1}{2})$  and  $2d(\frac{1}{2},0,0)$ , respectively. During the refinement, we used the constraint  $\text{occ}(\text{Ni}) + \text{occ}(\text{Os}) = 1$  at the  $2c$  and  $2d$  sites. The refinements indicated that the sample is nonstoichiometric compared to the ideal composition  $\text{La}_2\text{NiOsO}_6$  and the following occupancies were found:  $\text{occ}(\text{Ni}1)/\text{occ}(\text{Os}1) = 0.187(3)/0.813(3)$  at  $2c$ , and  $\text{occ}(\text{Ni}2)/\text{occ}(\text{Os}2) = 1.001(2)/-0.001(2)$  at  $2d$ . This suggests that the  $2d$  site is fully occupied by Ni, and therefore in the final refinement, the Ni occupancy at the  $2d$  site was fixed to be 1. This gives the composition  $\text{La}_2\text{Ni}_{1.187(3)}\text{Os}_{0.813(3)}\text{O}_6$ . The refinements resulted in a residual  $R_F = 0.0180$  (defined as  $R_F = \sum \|F_{\text{obs}}\| - |F_{\text{calc}}| / \sum |F_{\text{obs}}|$ ). The sample contains a smaller amount of  $\text{La}_3\text{OsO}_7$  (5.6%) [25,26]. For this compound, only the overall scale factor and the lattice parameters were simultaneously allowed to vary during the Rietveld refinements, whereas the structural parameters, taken from Ref. [25], were fixed. The loss of Os is presumably due

TABLE I. Crystallographic data obtained from refinements of powder x-ray and neutron diffraction data of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  at different temperatures.

	295 K	160 K	3.2 K
Diffraction source	X ray	Neutron	Neutron
Space group	$P2_1/n$	$P2_1/n$	$P2_1/n$
Lattice parameters	$a = 5.58129(8) \text{ \AA}$ $b = 5.61271(9) \text{ \AA}$ $c = 7.89614(12) \text{ \AA}$ $\beta = 90.075(4)^\circ$	$a = 5.5741(5) \text{ \AA}$ $b = 5.6118(5) \text{ \AA}$ $c = 7.8902(7) \text{ \AA}$ $\beta = 90.02(4)^\circ$	$a = 5.5698(4) \text{ \AA}$ $b = 5.6123(4) \text{ \AA}$ $c = 7.8862(6) \text{ \AA}$ $\beta = 90.04(2)^\circ$
Cell volume	$247.356(7) \text{ \AA}^3$	$246.81(4) \text{ \AA}^3$	$246.52(3)$
$R_F$	0.0180	0.0250	0.0254

to the formation of volatile  $\text{OsO}_4$  during the synthesis. The EDX analysis revealed that the sample contains much more Ni than Os,  $\text{Ni}/\text{Os} = 1.17/0.83$ , which is close to the results from the Rietveld refinement. It is noted that a small amount of  $\text{La}_2\text{O}_3$  left in the sample can be seen by detailed inspection of the x-ray diffraction (XRD) pattern which balances the loss of Os.

Due to the fact that the x-ray scattering power of the O atoms is relatively weak in comparison to those of the much heavier La and Os atoms, the crystal structure was also investigated by neutron diffraction at 3.2 and 160 K, well below and above the magnetic ordering temperature of about

125 K. The crystal structure of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  could be successfully refined in the space group  $P2_1/n$  with residuals  $R_F = 0.0273$  (3.2 K) and  $R_F = 0.0250$  (160 K), respectively (Fig. 1, bottom). During the refinement, the occupancies of the Ni and Os atoms at the  $2c$  and  $2d$  sites were fixed to the values obtained from the x-ray data. Concerning the impurity phase  $\text{La}_3\text{OsO}_7$ , only the overall scale factor and the lattice parameters were simultaneously allowed to vary during the Rietveld refinements, whereas the structural parameters obtained at lower temperature (taken from Ref. [26]) were fixed. The obtained crystallographic data are summarized

TABLE II. The atomic positions and isotropic temperature factors ( $B_{\text{iso}}$ ) of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$ . The refinements were carried out in the monoclinic space group  $P2_1/n$ . The  $2c$  sites ( $\frac{1}{2}, 0, \frac{1}{2}$ ) are occupied by Ni1/Os1 atoms while the  $2d$  sites ( $0, 0, \frac{1}{2}$ ) are occupied by Ni2 atoms with the occupancy fixed to be 1. For the x-ray diffraction data the  $B_{\text{iso}}$  were constrained to be equal for the metal atoms and the oxygen atoms, respectively. For the neutron diffraction data the  $B_{\text{iso}}$  of the two metal sites were allowed to vary separately. Also shown are the magnetic moments obtained from data collected on E2. The  $2d$  site is fully occupied with Ni, while the  $2c$  site contains 19% Ni. For the Ni atoms at both sites we used the constraint  $\mu(\text{Ni1}) = \mu(\text{Ni2})$ , while the Os moment at the  $2c$  site is assumed to be zero. Thus for the averaged moment at site  $2c$  we used the constraint  $\mu(\text{Os1}/\text{Ni1, site } 2c) = 0.19 \mu(\text{Ni2, site } 2d)$ . All parameters marked with an asterisk were not allowed to vary during the refinements.

$T$ (K)	295	160	3.2
Diffraction source	X ray	Neutron	Neutron
$x$ (La)	$-0.0060(5)$	$0.0146(7)$	$0.0154(6)$
$y$ (La)	$0.0379(1)$	$0.0436(3)$	$0.0438(3)$
$z$ (La)	$0.2521(3)$	$0.2504(19)$	$0.2494(18)$
$x$ (O1)	$0.270(3)$	$0.2845(17)$	$0.2823(13)$
$y$ (O1)	$0.295(3)$	$0.2809(13)$	$0.2872(12)$
$z$ (O1)	$0.031(5)$	$0.0465(11)$	$0.0470(10)$
$x$ (O2)	$0.213(3)$	$0.1976(17)$	$0.1955(13)$
$y$ (O2)	$0.796(2)$	$0.7999(12)$	$0.7950(10)$
$z$ (O2)	$0.026(5)$	$0.0352(11)$	$0.0470(10)$
$x$ (O3)	$0.904(3)$	$0.9224(11)$	$0.9224(10)$
$y$ (O3)	$0.481(1)$	$0.4871(9)$	$0.4835(8)$
$z$ (O3)	$0.245^*$	$0.2478(12)$	$0.2475(10)$
$B_{\text{iso}}$ (La)	$0.16(2)$	$0.56(4)$	$0.43(4)$
$B_{\text{iso}}$ (Ni/Os)	$0.16(2)$	$0.29(2)$	$0.20(2)$
$B_{\text{iso}}$ (O)	$1^*$	$0.53(3)$	$0.43(3)$
occ (Ni1/Os1)	$0.187(3)/0.813(3)$	$0.187/0.813^*$	$0.187/0.813^*$
$\mu_x$ (Ni2, site $2d$ ) [ $\mu_B$ ]			$1.77(5)$
$\mu_x$ (Os1/Ni1, site $2c$ ) [ $\mu_B$ ]			$0.34(1)$
$\mu_y$ (Ni2, site $2d$ ) [ $\mu_B$ ]			$0.60(12)$
$\mu_y$ (Os1/Ni1, site $2c$ ) [ $\mu_B$ ]			$0.11(2)$
$\mu_{\text{exp}}$ (Ni)			$1.87(5)$

TABLE III. Selected bond lengths and bond angles of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  derived from x-ray and neutron powder patterns at different temperatures.

Bond length (Å)	295 K	160 K	3.2 K
Ni-O1	$2.109(16) \times 2$	$2.016(8) \times 2$	$2.051(7) \times 2$
Ni-O2	$1.981(16) \times 2$	$2.044(9) \times 2$	$2.083(7) \times 2$
Ni-O3	$2.086(5) \times 2$	$2.038(13) \times 2$	$2.039(9) \times 2$
Os-O1	$1.912(16) \times 2$	$2.040(9) \times 2$	$2.009(7) \times 2$
Os-O2	$2.052(15) \times 2$	$2.030(8) \times 2$	$2.016(7) \times 2$
Os-O3	$2.011(5) \times 2$	$2.004(10) \times 2$	$2.002(9) \times 2$
Ni-O1-Os	159.6(6)	154.4(3)	153.7(3)
Ni-O2-Os	157.8(6)	152.1(3)	149.4(3)
Ni-O3-Os	149.1(5)	154.9(2)	154.7(2)

in Table I, and the atomic positions are summarized in Table II.

The interoctahedral Ni-O-Os bond angles of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  deviate strongly from  $180^\circ$  (Table III), indicating strong octahedral tilting as shown in Fig. 2. The bond distances in the  $\text{NiO}_6$  and  $\text{OsO}_6$  octahedra are summarized in Table III. At 295 K, the average Ni-O bond length is 2.06 Å, which is comparable to the literature values of 2.05 and 2.08 Å for  $\text{Ni}^{2+}$  double perovskite oxides  $\text{A}_2\text{NiOsO}_6$  ( $\text{A} = \text{Ca}, \text{Ba}$ ) [6,27], respectively, indicating the divalent Ni in this sample. The bond valence sum calculated for Ni-centered octahedra [28,29] is 2.20, which is close to the values for typical  $\text{Ni}^{2+}$  double perovskites [30].

### B. X-ray absorption spectroscopy

To explore the valence state of Os, the Os-L<sub>3</sub> XAS spectrum of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  was recorded at room temperature. It is well known that XAS spectra are highly sensitive to the valence state: An increase of the valence state of the metal ion by 1 causes a shift of the XAS L<sub>2,3</sub> spectra by  $\geq 1$  eV toward higher energies [31,32]. Figure 3 shows the Os-L<sub>3</sub> XAS spectrum of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  together with spectra of  $\text{Sr}_2\text{FeOsO}_6$  and  $\text{La}_2\text{MgOsO}_6$  as  $\text{Os}^{5+}$  and  $\text{Os}^{4+}$  references, respectively [6]. The energy position of the  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  spectrum is located exactly in the middle between that of

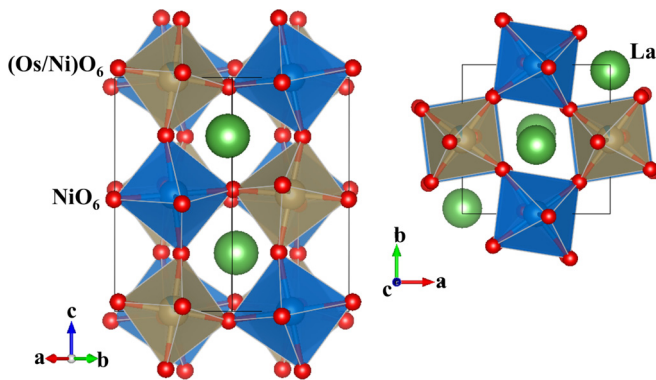


FIG. 2. Crystal structure of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$ . Blue octahedra are fully centered by  $\text{Ni}^{2+}$  ions; gray octahedra contain a majority fraction of Os ( $\text{Os}^{4+}$ ,  $\text{Os}^{5+}$ ) ions and a minor fraction of  $\text{Ni}^{2+}$  ions.

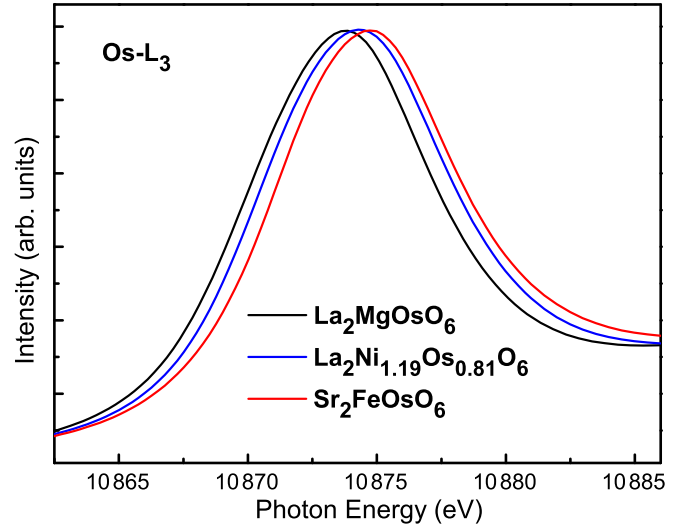


FIG. 3. The Os-L<sub>3</sub> XAS spectra of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  and of  $\text{Sr}_2\text{FeOsO}_6$  as  $\text{Os}^{5+}$  reference and  $\text{La}_2\text{MgOsO}_6$  as  $\text{Os}^{4+}$  reference.

$\text{Sr}_2\text{FeOsO}_6$  and  $\text{La}_2\text{MgOsO}_6$  demonstrating the oxidation state of  $\text{Os}^{4.5+}$ . This is consistent with the expected valence state  $\text{Os}^{4.46+}$  for the composition  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  obtained from the Rietveld refinements of the XRD data.

### C. Electrical transport

The electrical resistivity of a sintered polycrystalline sample of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  is displayed in Fig. 4. The sample shows insulating behavior as the resistivity increases by several orders of magnitude as the temperature decreases and exceeds the measurement limit for temperatures lower than 100 K. The data were plotted on  $T^{-1}$  and  $T^{-1/4}$  scales, and the plot is found to be roughly linear on a  $T^{-1/4}$  scale (see the inset of Fig. 4), in accordance with a three-dimensional variable range hopping transport model.

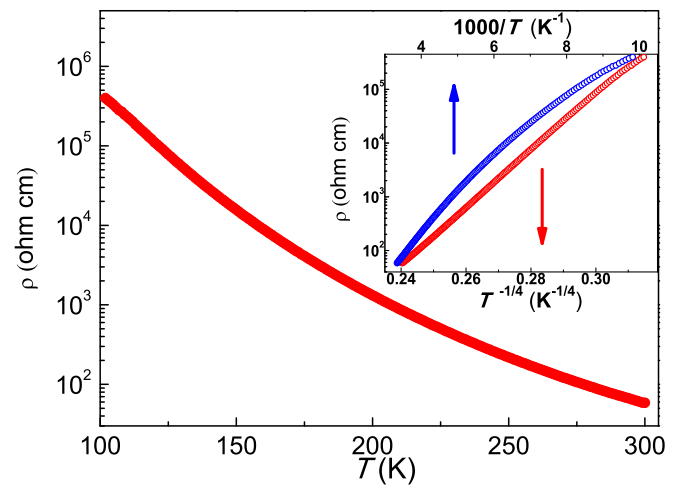


FIG. 4. Temperature dependence of the electrical resistivity of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$ . The inset shows the corresponding plots on  $T^{-1}$  and  $T^{-1/4}$  scales.



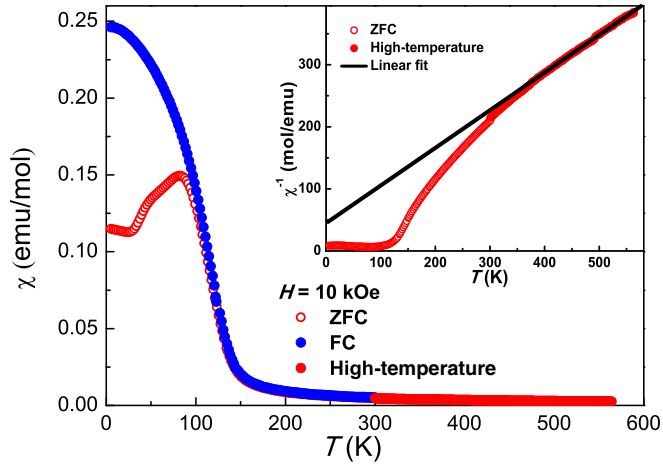


FIG. 5. Temperature dependence of the magnetic susceptibility of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$ . The inset shows the corresponding  $\chi^{-1}$  vs  $T$  plot.

#### D. Magnetic properties

The temperature dependence of the magnetic susceptibility  $\chi$  of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  is shown in Fig. 5. The sharp increase of  $\chi$  below about 125 K with cooling indicates a possible ferro- or ferrimagnetic transition. Because the minor impurity  $\text{La}_3\text{OsO}_7$  is antiferromagnetic with a Néel temperature of 45 K [26], the sharp transition around 125 K is intrinsic for  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$ . The convex  $\chi^{-1}$  vs  $T$  curve above the magnetic transition temperature indicates a ferrimagnetic transition. An attempt to fit the data above 400 K with the Curie-Weiss law resulted in a Curie-Weiss temperature ( $\theta_{\text{CW}}$ ) of  $-73$  K and an effective magnetic moment ( $\mu_{\text{eff}}$ ) per formula unit (f.u.) of  $3.63 \mu_{\text{B}}$ . The negative  $\theta_{\text{CW}}$  indicates that antiferromagnetic interactions are dominant in this sample, supporting the ferrimagnetic phase transition. The obtained  $\mu_{\text{eff}}$  compares reasonably well with  $\mu_{\text{eff}} = 3.69 \mu_{\text{B}}/\text{f.u.}$ , which is calculated from the formula  $\text{La}_2\text{Ni}_{1.19}^{2+}\text{Os}_{0.43}^{4+}\text{Os}_{0.38}^{5+}\text{O}_6$  if one assumes a spin-only moment for  $\text{Ni}^{2+}$  ( $2.83 \mu_{\text{B}}$ ), and takes the average

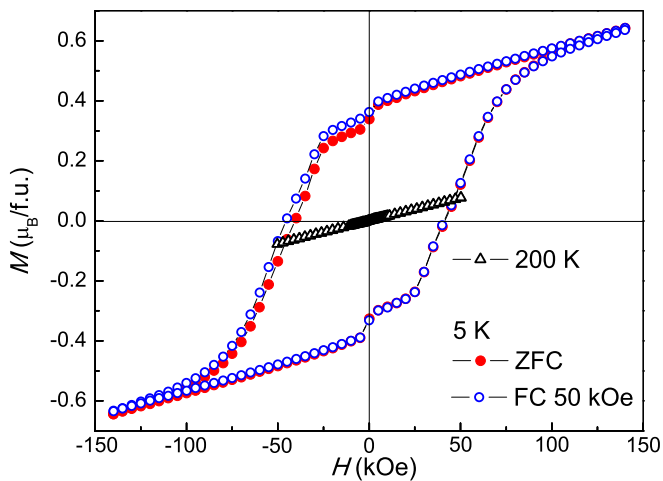


FIG. 6. The isothermal magnetization of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  measured at 200 and 5 K. The small step near  $H = 0$  may be due to a soft magnetic impurity.

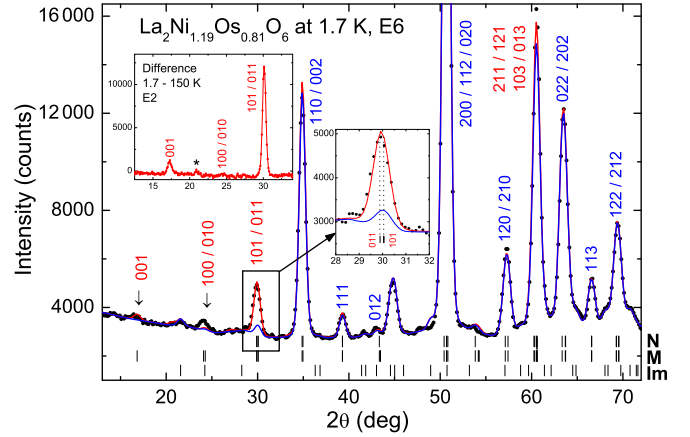


FIG. 7. Neutron powder pattern of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  taken on E6 at 1.7 K. The calculated patterns [nuclear contribution (blue), sum of the nuclear and the magnetic contribution (red)] are compared with the observations (black-filled circles). The positions (black bars) of the nuclear (N) and magnetic (M) Bragg reflections of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$ , and those of the impurity phase  $\text{La}_3\text{OsO}_7$  (Im) are shown. The region where the strongest magnetic reflection  $(011)_{\text{M}}$ , and the pair  $(101)_{\text{M}}$  and  $(10-1)_{\text{M}}$  occur is shown in the inset. The second inset (left) shows the presence of the magnetic reflection  $(001)_{\text{M}}$  as obtained from the difference pattern (1.7–150 K) from E2 data. The weak magnetic intensity observed at  $21.5^\circ$  (marked with an asterisk) can be assigned to the magnetic reflection  $(\frac{1}{2}, \frac{1}{2}, 1)_{\text{M}}$  of the impurity  $\text{La}_3\text{OsO}_7$  [26]. All prominent magnetic reflections are labeled in red.

$\mu_{\text{eff}} = 3.28 \mu_{\text{B}}$  for  $\text{Os}^{5+}$  as obtained for some  $\text{Os}^{5+}$  double perovskites [33–36]. Any possible Van-Vleck type paramagnetic contribution to  $\mu_{\text{eff}}$  of the  $\text{Os}^{4+}$  ions is neglected in this rough estimate.

To further characterize the magnetic transition, the isothermal magnetization curves of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  were measured above and below the transition temperature after zero-field cooling. The linear behavior of the  $M(H)$  data at 200 K is consistent with the paramagnetic state. At 5 K, the  $M(H)$  curve collected between  $-50$  and  $+50$  kOe shows a large hysteresis, but the hysteresis loop cannot be closed. Thus, the  $M(H)$  curve was measured between  $-140$  and  $+140$  kOe; see Fig. 6. The magnetization is about  $0.65 \mu_{\text{B}}/\text{f.u.}$  at 5 K and 140 kOe, but still not saturated. A remarkable feature of the  $M(H)$  curve of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  is the high coercive field,  $H_{\text{C}} = 41$  kOe, at  $T = 5$  K. The  $M(H)$  curve was also measured after field cooling (50 kOe) and found to be identical to that measured after zero-field cooling; no exchange bias effect is observed.

#### E. Magnetic structure

In order to investigate the magnetic structure of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$ , we have collected several neutron powder patterns on the E6 instrument of BER II ( $\lambda = 2.42 \text{ \AA}$ ) from 1.7 up to 134 K. In agreement with our magnetization measurements a magnetic contribution to the powder patterns could be observed below 125 K. The strongest magnetic intensity was observed at the position of the reflections 011, 101, and  $10-1$  (Fig. 7). For the Ni and Os atoms at the positions  $2d(\frac{1}{2}, 0, 0; 0, \frac{1}{2}, \frac{1}{2})$  and  $2c(\frac{1}{2}, 0, \frac{1}{2}; 0, \frac{1}{2}, 0)$  magnetic intensity could be generated with a model, where the spins located

at both sites are coupled antiparallel with a spin sequence  $- - + +$ . In the pattern, shown in Fig. 7, additional weak intensities were observed at  $2\theta = 16.7^\circ$  and  $24.1^\circ$ , which are close to the position of the reflection 001, and that of the reflection pair 100/010 (forbidden in  $P2_1/n$ ), respectively. The third reflection observed at  $21.5^\circ$  can be indexed as  $(\frac{1}{2}, \frac{1}{2}, 1)_M$  belonging to the impurity  $\text{La}_3\text{OsO}_7$  [26]. In order to characterize these reflections in more detail, we have collected powder patterns at 1.7 and 150 K on the E2 instrument with a much higher counting rate and better instrumental resolution. From the difference pattern shown in the inset of Fig. 7 magnetic intensity was clearly detected for reflection 001, whereas for the reflection pair 100 and 010 the magnetic contribution was found to be much weaker. Therefore the intensity observed at  $24.1^\circ$  should belong to another unknown impurity, which does not change its intensity from 1.7 to 150 K.

From our x-ray diffraction study, it was found that the osmium  $2c$  site is partially occupied with Ni atoms giving a ratio  $\text{occ}(\text{Ni}1)/\text{occ}(\text{Os}1) = 0.187(3)/0.813(3)$ , while the  $2d$  site is fully occupied by Ni2. For the determination of the moment values from the refinements of the magnetic structure we have assumed that Ni1 and Ni2, located at the  $2c$  and  $2d$  sites, have the same magnetic moment using the constraint  $\mu(\text{Ni}1) = \mu(\text{Ni}2)$ , while initially the Os moments were kept at zero. Further, we used the positional and thermal parameters as well as the monoclinic angle  $\beta$  as determined more precisely from the analysis of the E9 data. In order to determine the moment direction, we carried out Rietveld refinements using model structures. In the case of an ordering along the  $b$  and  $c$  axes, it could be seen that the magnetic peak significantly shifted to a higher  $2\theta$  value, while a satisfactory fit was obtained when the moments are aligned to the  $a$  axis. In fact, due to the lattice distortions the reflection pair  $101/10-1$  is shifted to a higher  $2\theta$  value in comparison to the 011 (see inset of Fig. 7). From our calculations, we obtained the following intensity ratios:  $I(011/01-1)/I(101/10-1): \sim 3/1$  ( $\mu//a$ ),  $\sim 1/3$  ( $\mu//b$ ),  $\sim 1/1$  ( $\mu//c$ ). This shows that the best calculated peak position is reached when the strongest magnetic intensity appears on the reflection 011 generated with a moment direction parallel to the  $a$  axis. In the next step, we allowed varying the magnetic moment of the Os atoms at the  $2c$  site. Independently from the models used above the refined Os moment was found to be  $0.00(6)\mu_B$ . Therefore in the final refinement, the Os moment was fixed to be zero, with the consequence that only the Ni1 moments at the mixed  $2c$  site are coupled antiparallel to the Ni2 moments at the  $2c$  site. Due to the fact that the mixed site is occupied with 19% Ni we used the constraint  $\mu(\text{Os}1/\text{Ni}1, 2c \text{ site}) = 0.19\mu(\text{Ni}2, 2d \text{ site})$ .

In order to generate magnetic intensity at the position of the reflection 001 two models were considered, where the Ni and mixed Os/Ni sites  $(\frac{1}{2}, 0, 0)$ ,  $(0, \frac{1}{2}, \frac{1}{2})$ ,  $(\frac{1}{2}, 0, \frac{1}{2})$ , and  $(0, \frac{1}{2}, 0)$  have an additional  $b$  spin component with the sequences  $+ - - +$  and  $+ - + -$ , respectively. For the refinements we have used the same constraints as given above. The use of the spin sequence  $+ - + -$  results in calculated intensities for the reflections 100 and 001 with the ratio  $I(100)/I(001) \sim 1/1$ . However, the inset of Fig. 7 shows that the magnetic intensity of the 100 is much weaker than that of the 001 reflection. A reasonable fit was obtained by using the spin sequence  $+ - - +$ , where the calculated ratio was found to

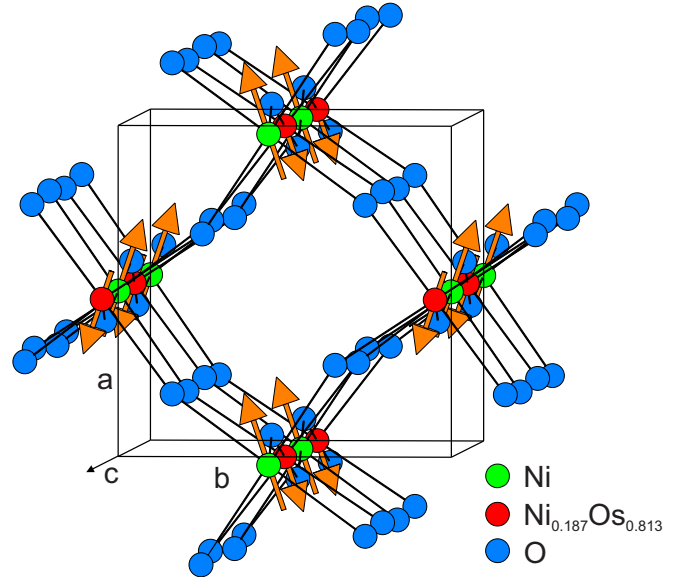


FIG. 8. Magnetic structure of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$ . For the Ni and mixed Os/Ni sites at  $2d(\frac{1}{2}, 0, 0; 0, \frac{1}{2}, \frac{1}{2})$  and  $2c(\frac{1}{2}, 0, \frac{1}{2}; 0, \frac{1}{2}, 0)$  only the Ni atoms contribute to the magnetic ordering. The Os site was found to be partially occupied with 19% Ni. The Ni moments at both sites were constrained to be equal during the Rietveld refinements resulting in a moment  $\mu_{\text{exp}}(\text{Ni}) = 1.90(5)\mu_B$ . Therefore the total moment at the Os/Ni site only reached the value  $0.19 \times \mu_{\text{exp}}(\text{Ni}) = 0.36(1)\mu_B$ . For clarity, the magnitude of the Ni moment at the Os/Ni site is exaggerated. The magnetic atoms form ferrimagnetic chains along the  $c$  axis, and within the  $ab$  plane one finds a noncollinear spin arrangement.

be  $I(100)/I(001) \sim 1/4.5$ . Together with the spin component along the  $a$  direction with the spin sequence  $+ + - -$  one obtains a spin structure which is noncollinear in the  $ab$  plane, but has collinear spin chains along the  $c$  direction (Fig. 8). This spin arrangement with the propagation vector  $\mathbf{k} = 0$  is compatible with the crystal structure symmetry of the space group  $P2_1/n$  as shown by a symmetry analysis [37]. For this analysis we used the program BASIREP of the FULLPROF suite [21]. For each site,  $2d(\frac{1}{2}, 0, 0; 0, \frac{1}{2}, \frac{1}{2})$  and  $2c(\frac{1}{2}, 0, \frac{1}{2}; 0, \frac{1}{2}, 0)$ , one finds the two irreducible representations:  $\Gamma_1$  with the modes  $A_x(+ -), F_y(+ +), A_z(+ -)$ , and  $\Gamma_2$  with  $F_x(+ +), A_y(+ -), F_z(+ +)$ . Our observation is compatible with  $\Gamma_2$ , where each sublattice shows the spin sequence  $+ -$  along the  $b$  direction and  $+ +$  along  $a$  and  $c$ . Due to the ferrimagnetic spin alignment along the  $a$  axis one finds in total the sequence  $+ +, - -$  along this direction, and along the  $b$  axis the sequence  $+ -, - +$ , as already given above. At this point it is noted that the symmetry analysis also allows a ferrimagnetic spin alignment along the  $c$  axis. Therefore in the refinement we additionally allowed to vary the Ni moment along  $c$ . This resulted in a not well defined value  $\mu_z(\text{Ni}) = 0.2(4)\mu_B$ , which is close to zero. Therefore we conclude that a magnetic ordering practically occurs in the  $ab$  plane as shown in Fig. 8. Finally for the 1.7 K data set collected on E2 we obtained the moment values  $\mu_x(\text{Ni}) = 1.77(5)\mu_B$  and  $\mu_y(\text{Ni}) = 0.60(12)\mu_B$ , resulting in a total moment  $\mu_{\text{exp}}(\text{Ni}) = 1.87(5)\mu_B$  and a residual  $R_M = 0.073$  (defined as  $R_M = \sum \|I_{\text{obs}}\| - \|I_{\text{calc}}\| / \sum \|I_{\text{obs}}\|$ ). In this

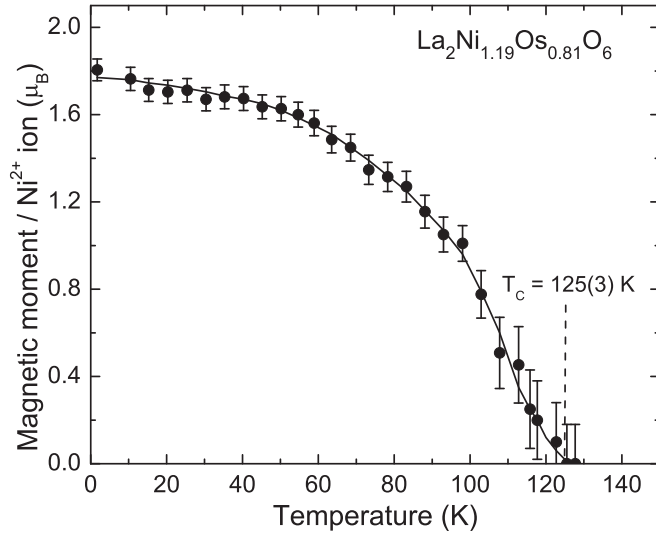


FIG. 9. Temperature dependence of the magnetic moments of the  $\text{Ni}^{2+}$  ions in  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$ . The Wyckoff  $2c$  site is occupied with 81% Os1 and 19% Ni1, while the  $2d$  site only contains Ni2. During the refinements of the magnetic moments, we have applied the constraint  $\mu(\text{Ni1}) = \mu(\text{Ni2})$ , while the magnetic moment of Os was set to be zero. The bold line is a guide for the eye.

noncollinear magnetic structure, the moments form a tilting angle to the  $a$  axis of  $19(1)^\circ$ . For the Rietveld refinements of the powder patterns collected on E6, we have fixed the component  $\mu_y(\text{Ni})$ . Here we have obtained the component  $\mu_x(\text{Ni}) = 1.81(5) \mu_B$  and a total moment  $\mu_{\text{exp}}(\text{Ni}) = 1.90(5) \mu_B$ , respectively. This value is only slightly smaller than the spin-only moment of  $2.0 \mu_B$  expected for  $\text{Ni}^{2+}$  ions in an octahedral coordination environment ( $t_{2g}^6 e_g^2$  electron configuration). Due to the fact that the mixed Os/Ni site contains 19% Ni the total moment is  $\mu_x(\text{Os/Ni}) = 0.34 \mu_B$ . Due to the ferrimagnetic coupling along  $a$  the resultant ferromagnetic component is reduced to  $1.4\text{--}1.5 \mu_B$ . The temperature dependence of the obtained Ni moment, shown in Fig. 9, yields an ordering temperature of  $125(3)$  K. Due to the weakness of the magnetic reflection  $(001)_M$  we only could follow the temperature dependence of the moment of the much stronger pronounced  $x$  component from the E6 data.

Our data analysis suggests that the observed ferrimagnetic structure is driven by the antiferromagnetic interactions between  $\text{Ni}^{2+}$  ions on the  $2d$  sites and the additional  $\text{Ni}^{2+}$  ions

on the  $2c$  sites, which leads to a partial cancellation of magnetic moments. Although the obtained  $\text{Ni}^{2+}$  moment is reasonable it is noted that an unambiguous refinement of Os moments in double perovskites with two magnetic ions at the  $B$  and  $B'$  site is difficult [5,30] and thus it cannot be ruled out that ordered  $\text{Os}^{5+}$  moments contribute to the stabilization of the ferrimagnetic spin structure.

#### IV. DISCUSSION

We synthesized a nonstoichiometric double perovskite  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  containing merely  $\text{Ni}^{2+}$  ions at the  $2d$  sites but a mixed occupancy of paramagnetic ( $\text{Ni}^{2+}$ ,  $\text{Os}^{5+}$ ) and nonmagnetic ( $\text{Os}^{4+}$ ) ions at the  $2c$  sites of the monoclinic crystal structure. Formally the compound may be written as  $\text{La}_2\text{Ni}^{2+}(\text{Ni}_{0.19}^{2+}\text{Os}_{0.43}^{4+}\text{Os}_{0.38}^{5+})\text{O}_6$ . The most remarkable property of this insulating compound is that it features a ferrimagnetic transition with an extraordinarily broad hysteresis at 5 K. The spin structure is noncollinear in the  $ab$  plane.

For rationalizing the magnetic properties we consider first the stoichiometric double perovskite  $\text{La}_2\text{Ni}^{2+}\text{Os}^{4+}\text{O}_6$  with the perfect atomic order, where the magnetism should be entirely determined by antiferromagnetic exchange interactions between the half-filled  $e_g$  orbitals of the  $\text{Ni}^{2+}$  ions as the  $\text{Os}^{4+}$  ions are expected to be nonmagnetic ( $J_{\text{eff}} = 0$ ). Therefore, for  $\text{La}_2\text{NiOsO}_6$  antiferromagnetic ordering is anticipated as is indeed observed for other  $A_2\text{NiB}'\text{O}_6$  compounds with nonmagnetic ions on the  $B'$  site ( $B' = \text{W}^{6+}, \text{Ti}^{4+}, \text{Ir}^{5+}$ ) [38–41], Table IV. The adopted antiferromagnetic spin structure will be determined mainly by the balance between the nearest neighbor  $\text{Ni}^{2+}\text{--O--O--Ni}^{2+}$  interactions and the next nearest neighbor  $\text{Ni}^{2+}\text{--O--Os}^{4+}\text{--O--Ni}^{2+}$  exchange pathways. However, due to the nonstoichiometry the actual compound  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  contains a large fraction of paramagnetic  $\text{Ni}^{2+}$  and  $\text{Os}^{5+}$  ions at the  $2c$  sites, which is the origin for the observed ferrimagnetism with  $T_C = 125$  K. Our analysis of the powder neutron diffraction data indicates that the ferrimagnetism may be solely driven by the antiferromagnetic interactions between the majority of  $\text{Ni}^{2+}$  ions at the  $2d$  sites and the minority of  $\text{Ni}^{2+}$  ions at the  $2c$  sites, which implies that the  $\text{Os}^{5+}$  moments remain magnetically disordered and possibly freeze at low temperatures. The strong  $\text{Ni}^{2+}\text{--O--Ni}^{2+}$   $e_g(\sigma) - e_g(\sigma)$  exchange pathways are possibly the origin for the enhanced ordering temperature of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  compared to other  $\text{Ni}^{2+}$  double perovskites without these pathways (Table IV). However, due to the above-mentioned

TABLE IV. Comparison of the magnetic properties of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  with those of some  $\text{Ni}^{2+}$  based double perovskite oxides. The ions on the  $B'$  site of the double perovskite structure are either nonmagnetic ( $\text{W}^{6+}$ ,  $\text{Ti}^{4+}$ ,  $\text{Ir}^{5+}$ ) or magnetic ( $\text{Os}^{5+}$ ,  $\text{Ir}^{6+}$ ).

Magnetic ions	Composition	Space group	Properties	$T_m$ (K)	$\theta_w$ (K)	$\mu_{\text{eff}} (\mu_B)$	Reference
$\text{Ni}^{2+}$	$\text{Ca}_2\text{NiWO}_6$	$P2_1/n$	AFM	52.5	−75	2.85	[38]
$\text{Ni}^{2+}$	$\text{Sr}_2\text{NiWO}_6$	$I4/m$	AFM	54	−175	3.05	[39]
$\text{Ni}^{2+}$	$\text{La}_2\text{NiTiO}_6$	$P2_1/n$	AFM	25	−60	3.12	[40]
$\text{Ni}^{2+}$	$\text{SrLaNiIrO}_6$	$P2_1/n$	AFM	74	−90	3.3	[41]
$\text{Ni}^{2+} + \text{Os}^{4.5+}$	$\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$	$P2_1/n$	FIM	125	−73	3.63	This work
$\text{Ni}^{2+} + \text{Os}^{5+}$	$\text{SrLaNiOsO}_6$	$P2_1/n$	AFM	60	−23	4.13	[42]
$\text{Ni}^{2+} + \text{Os}^{5+}$	$\text{CaLaNiOsO}_6$	$P2_1/n$	AFM	30	−83	3.87	[30]
$\text{Ni}^{2+} + \text{Ir}^{6+}$	$\text{Sr}_2\text{NiIrO}_6$	$P2_1/n$	AFM	58			[43]

difficulties in resolving the Os moment from powder neutron data this analysis may not be unambiguous and also antiferromagnetic  $\text{Ni}^{2+}\text{--O--Os}^{5+}$  interactions may contribute to the stabilization of the noncollinear ferrimagnetic spin structure. The latter interactions are a consequence of the distorted crystal structure. Since the  $t_{2g}$  orbitals of  $\text{Ni}^{2+}$  are completely filled, the only possible antiferromagnetic nearest neighbor exchange pathway is via virtual hopping between half-filled  $\text{Ni-}e_g$  and  $\text{Os-}t_{2g}$  orbitals. The  $\text{Ni-}e_g$  and  $\text{Os-}t_{2g}$  orbitals are orthogonal in the cubic double perovskite structure, but these interactions become possible in the monoclinic double perovskite where the  $\text{Ni--O--Os}$  bond angles are strongly reduced from  $180^\circ$ . This mechanism was invoked for explaining the ferrimagnetic state in  $\text{Ca}_2\text{Ni}^{2+}\text{Os}^{6+}\text{O}_6$  [44]. Antiferromagnetic interactions between  $\text{Ni}^{2+}$  ( $d^8$ ) and  $\text{Os}^{5+}$  ( $d^3$ ) ions appear to be in contradiction to the Goodenough-Kanamori rules which predict a ferromagnetic interaction due to virtual hopping between the half-filled and empty  $e_g$  orbitals at the  $\text{Ni}^{2+}$  and  $\text{Os}^{5+}$  sites, respectively. However, compared to pure  $3d$  systems like  $\text{La}_2\text{NiMnO}_6$  [45,46] the crystal field splitting at the  $5d$  sites is much enhanced and the ferromagnetic coupling involving the  $\sigma$ -exchange pathway between the  $e_g$  orbitals becomes weak, in particular in the distorted double perovskite structure [47,48].

Most remarkably the magnetization of the present powder sample of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  features a broad hysteresis with a coercive field as high as 41 kOe. Similar giant coercive fields have been reported for diverse materials classes. Examples include rare-earth transition-metal films [49], ferrimagnetic transition-metal-radical single chain magnets [50], and intermetallic systems like  $\text{Mn}_{1.5}\text{Ga}$  films [51], as well as ferrimagnetic  $\text{Fe}_3\text{Se}_4$  nanostructures [52]. In transition-metal oxides extremely high coercivities of 90 and 120 kOe were found for single crystals of the ferrimagnet  $\text{LuFe}_2\text{O}_4$  [53] and weakly ferromagnetic  $\text{Sr}_5\text{Ru}_{5-x}\text{O}_{15}$  [54], respectively. Such materials are of interest in the search of rare-earth-free hard magnets. The detailed mechanism of giant coercivity is not generally understood. However, large magnetocrystalline anisotropy in combination with frustration and/or defects and disorder appears to favor huge coercivities [53]. A large magnetocrystalline anisotropy is indeed expected for  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$

due to the low-symmetry crystal structure and the presence of the heavy Os atoms. The noncollinear spin arrangement in the  $ab$  plane can be attributed to the Dzyaloshinskii-Moriya interaction which induces spin canting, whereas the high coercivity is possibly associated with a peculiar microstructure and competing exchange interactions introduced by the atomic disorder.

## V. CONCLUSIONS

Targeting the  $\text{Os}^{4+}$  ( $J_{\text{eff}} = 0$ ) compound  $\text{La}_2\text{NiOsO}_6$ , we actually obtained nonstoichiometric polycrystalline samples of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  by solid-state reaction. As many other double perovskite oxides,  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  crystallizes in the monoclinic space group  $P2_1/n$ . Here, the  $B$  sites are fully occupied by  $\text{Ni}^{2+}$  ions whereas the  $B'$  sites feature a mixed occupancy with paramagnetic  $\text{Ni}^{2+}$  and  $\text{Os}^{5+}$  as well as nonmagnetic  $\text{Os}^{4+}$  ions. Whereas stoichiometric  $\text{La}_2\text{NiOsO}_6$  is expected to be an antiferromagnet,  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  reveals a ferrimagnetic transition at 125 K and adopts a ferrimagnetic spin arrangement with collinear chains along the  $c$  axis, but spin canting in the  $ab$  plane. Our data analysis indicates that only  $\text{Ni}^{2+}$  magnetic moments are long-range ordered, but the behavior of the  $\text{Os}^{5+}$  moments is not entirely clear. The magnetization curve of  $\text{La}_2\text{Ni}_{1.19}\text{Os}_{0.81}\text{O}_6$  is characterized by a huge coercive field,  $H_C = 41$  kOe at  $T = 5$  K, which is attributed to the combined influence of large magnetocrystalline anisotropy introduced by the Os ions having strong spin-orbit coupling and atomic disorder. Our results suggest that careful tuning of the chemical composition may lead to the discovery of rare-earth-free hard magnets in the class of double perovskite oxides.

## ACKNOWLEDGMENTS

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