

Unraveling the complex magnetic structure of multiferroic pyroxene $\text{NaFeGe}_2\text{O}_6$: A combined experimental and theoretical study

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Magnetic order and the underlying magnetic model of the multiferroic pyroxene $\text{NaFeGe}_2\text{O}_6$ are systematically investigated by neutron powder diffraction, thermodynamic measurements, density-functional band-structure calculations, and Monte Carlo simulations. Upon cooling, $\text{NaFeGe}_2\text{O}_6$ first reveals one-dimensional spin-spin correlations in the paramagnetic state below about 50 K, uncovered by magnetic diffuse scattering. The sinusoidal spin-density wave with spins along the a direction sets in at 13 K, followed by the cycloidal configuration with spins lying in the (ac) plane below 11.6 K. Microscopically, the strongest magnetic coupling runs along the structural chains, $J_1 \simeq 12$ K, which is likely related to the one-dimensional spin-spin correlations. The interchain couplings $J_2 \simeq 3.8$ K and $J_3 \simeq 2.1$ K are energetically well balanced and compete, thus giving rise to the incommensurate order, in sharp contrast to other transition-metal pyroxenes, in which one type of the interchain couplings prevails. The magnetic model of $\text{NaFeGe}_2\text{O}_6$ is further completed by the weak single-ion anisotropy along the a direction. Our results resolve the earlier controversies regarding the magnetic order in $\text{NaFeGe}_2\text{O}_6$ and establish relevant symmetries of the magnetic structures. These results, combined with symmetry analysis, enable us to identify the possible mechanisms of the magnetoelectric coupling in this compound. We also elucidate microscopic conditions for the formation of incommensurate magnetic order in pyroxenes.

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I. INTRODUCTION

Spin-driven multiferroics, where significant coupling between magnetic order and electric polarization emerges due to simultaneous symmetry breaking induced by the incommensurate magnetic structure [1], have drawn a great deal of attention in recent years. Several theoretical models, such as the inverse Dzyaloshinskii-Moriya (DM) (or spin-current) model [2,3] and spin-dependent p - d orbital hybridization [4], have been put forward to explain this fascinating phenomenon. Although the inverse DM model essentially captured the behavior of many multiferroic materials, such as TbMnO_3 [5] and AgFeO_2 [6] with the cycloidal spin configuration and propagation vector lying in the spin plane, it failed to account for multiferroicity in systems with proper-screw magnetic symmetry. More recently, the mechanism of ferroaxiality of the crystal structure was proposed to explain the experimentally observed multiferroic properties of $\text{Cu}_3\text{Nb}_2\text{O}_8$ [7], $\text{CaMn}_7\text{O}_{12}$ [8], and $\text{RbFe}(\text{MoO}_4)_2$ [9], for which proper-screw magnetic structures with the spin plane perpendicular to the propagation vector have been found. In addition, Kaplan and Mahanti [10] have shown that the extended inverse DM effect may contribute to the microscopic electric polarization in both cycloid and proper-screw helical

systems. This observation was used to account for the multiferroicity in some of the delafossites [11].

As one of the main components of the Earth's crust and upper mantle, pyroxenes with the chemical formula AMT_2O_6 (A = mono- or divalent metal, M = transition metal, T = Ge or Si) have gained renewed attention of condensed-matter physicists since, recently, a number of magnetic pyroxenes were found to show multiferroicity or the magnetoelectric effect [12–17]. Subsequent investigation showed that only $\text{NaFeGe}_2\text{O}_6$ [14], $\text{SrMnGe}_2\text{O}_6$ [16], and the mineral aegirine [12] are truly multiferroic.

$\text{NaFeGe}_2\text{O}_6$ crystallizes in the space group $C2/c1'$. The zigzag chains of edge-sharing FeO_6 octahedra are bridged by corner-linked GeO_4 tetrahedral chains (Fig. 1). This structural one-dimensionality gives rise to the broad maximum in the magnetic susceptibility around 35 K. Two consecutive magnetic transitions at $T_{N2} = 13$ K and $T_{N1} = 11.6$ K, respectively, were identified through the specific-heat measurements [18,19]. The second transition is accompanied by the formation of spontaneous electric polarization confirmed by electric polarization measurements on both powders and single crystals [14,15].

Neutron diffraction studies suggest the incommensurate (ICM) nature of the magnetic order in $\text{NaFeGe}_2\text{O}_6$ [18,19]. However, even the periodicity of the magnetic structure remains controversial. Two different propagation vectors, $\mathbf{k} = (0.3357, 0, 0.0814)$ [19] and $\mathbf{k} = (0.323, 1.0, 0.08)$ [18], were reported by different groups. These vectors cannot be transformed into each other because $(0,1,0)$ is not a reciprocal

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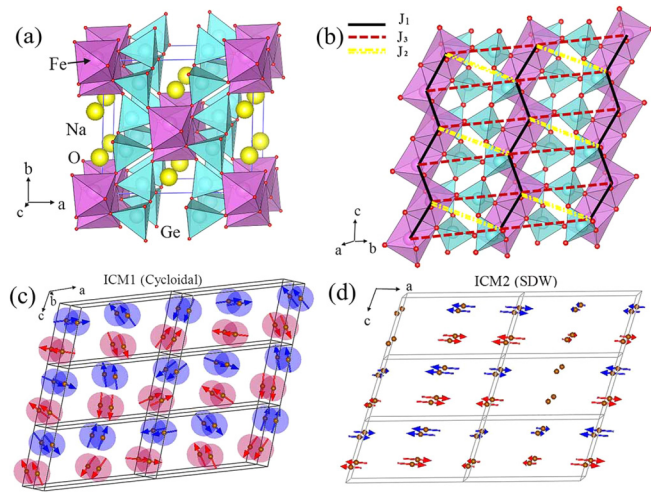


FIG. 1. (a) and (b) The crystal structure of NaFeGe₂O₆ with projections along different directions. The essential exchange interactions within and between the chains are highlighted. Schematic drawings of (c) the cycloidal spin configuration and (d) spin-density wave (SDW).

translation in the presence of C centering. The magnetic structure below T_{N1} was determined to be cycloidal, whereas the magnetic structure of the intermediate phase formed between T_{N1} and T_{N2} has not been reported to date. The controversial information on the magnetic structure, along with the absence of any established microscopic magnetic model, hinders further work on NaFeGe₂O₆ and curtails our understanding of the multiferroicity of this compound.

In the following, we revisit the magnetic structure of NaFeGe₂O₆ and establish the microscopic magnetic model. We demonstrate that the well-tuned balance between the interchain couplings gives rise to the incommensurate order and renders NaFeGe₂O₆ different from the majority of pyroxenes that feature collinear and commensurate magnetic structures. We further resolve the intermediate-temperature magnetic structure between T_{N1} and T_{N2} as the spin-density wave caused by the weak single-ion anisotropy of Fe³⁺. We finally discuss the implications of our results for the multiferroic behavior, as well as microscopic conditions for the formation of incommensurate magnetic order in pyroxenes.

II. METHODS

Polycrystalline NaFeGe₂O₆ was synthesized by a solid-state reaction. The stoichiometric mixture of reagent-grade Na₂CO₃, Fe₂O₃, and GeO₂ was ground in an agate mortar and pelletized. The pellets were placed into alumina crucibles and heated in air at 1273 K for 100 h and cooled down to room temperature. Intermediate regrinding and reheating were performed in order to improve the purity of the sample.

The temperature-dependent magnetic susceptibility was measured using a superconducting quantum interference device magnetometer (Quantum Design, MPMS-7T). The dc magnetic susceptibility was recorded from 2 to 350 K with zero-field-cooled and field-cooled procedures in a magnetic field of 1 T. The specific-heat measurement was carried out using a relaxation technique with a Quantum Design physical

property measurement system in the temperature range of 2–300 K on cooling. The pelletized sample was mounted on a sample platform with Apiezon N grease for better thermal contact.

Temperature-dependent powder x-ray diffraction (XRD) data were collected with a RIGAKU Smartlab diffractometer in the high-resolution parallel-beam mode using a Ge (220) \times 2 monochromator for Cu $K\alpha_1$ radiation and an Oxford Phenix cold stage, giving access to sample temperatures as low as 12 K. The neutron powder diffraction (NPD) data were collected at the ISIS pulsed neutron and muon facility of the Rutherford Appleton Laboratory (United Kingdom), on the WISH diffractometer located at the second target station [20]. A powder sample (~ 4.1 g) was loaded into a 6-mm-diameter cylindrical vanadium can and measured in the temperature range of 1.5–150 K using an Oxford Instrument cryostat. The data at 1.5, 20, 50, 100, and 150 K were collected for 1 h, and typical scans between these temperatures were carried out with an exposition time of 30 min with steps of 1 K in the temperature range of 2–10 K and 0.2 K for measurements between 10 and 15 K.

Rietveld refinements of the crystal and magnetic structures were performed using the FULLPROF program [21] against the data measured in the detector banks at average 2θ values of 58°, 90°, 122°, and 154°, each covering 32° of the scattering plane. Group-theoretical calculations were done using ISODISTORT [22] and Bilbao Crystallographic Server (Magnetic Symmetry and Applications [23]) software.

Magnetic exchange couplings were analyzed using density-functional theory (DFT) band-structure calculations performed in the FPLO [24] and VASP [25,26] codes. The Perdew-Burke-Ernzerhof flavor of the exchange-correlation potential was chosen [27]. A k mesh with up to 64 points in the symmetry-irreducible part of the first Brillouin zone was used and proved sufficient for the full convergence with respect to the number of k points. Correlation effects in the Fe 3d shell were taken into account on the mean-field level via the DFT+ U procedure with the on-site Coulomb repulsion $U_d = 6$ –8 eV and Hund's exchange $J_d = 1$ eV [28,29].

Exchange couplings J_{ij} enter the spin Hamiltonian

$$H = \sum_{\langle ij \rangle} J_{ij} \mathbf{S}_i \mathbf{S}_j + \sum_i A_i (S_i^z)^2, \quad (1)$$

where $S = \frac{5}{2}$ and the summation is over bonds $\langle ij \rangle$. The values of J_{ij} were obtained by a mapping procedure using energies of collinear spin configurations [30]. A similar mapping procedure for orthogonal spin configurations yields magnetic anisotropy parameters A_i when spin-orbit (SO) coupling is included within the DFT+ U +SO approach.

The thermodynamic properties of the resulting spin model were analyzed by classical Monte Carlo simulations using the SPINMC algorithm of the ALPS package [31]. Finite $L \times L \times L$ lattices with $L \leq 8$ and periodic boundary conditions were used.

III. RESULTS

A. Magnetic properties

The temperature dependence of the magnetic susceptibility of NaFeGe₂O₆ measured in a magnetic field of 1 T is shown

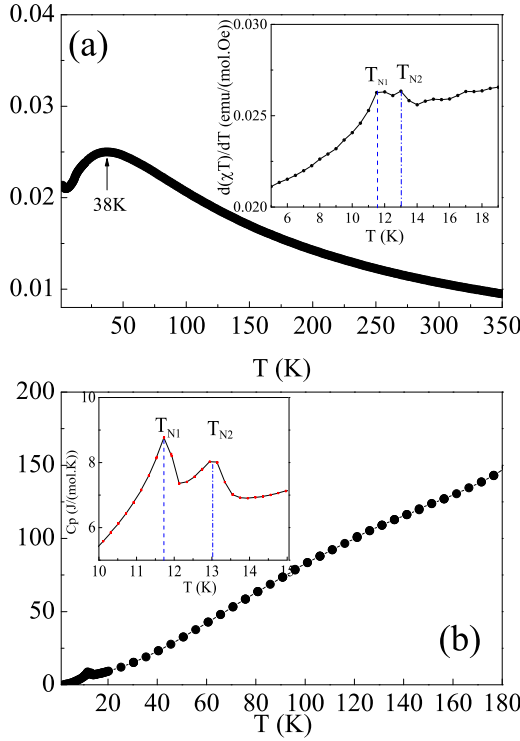


FIG. 2. (a) Temperature dependence of the magnetic susceptibility of $\text{NaFeGe}_2\text{O}_6$ in a magnetic field $H = 1$ T. The inset shows Fisher's heat capacity $d(\chi T)/dT$. (b) Heat capacity C_p for $\text{NaFeGe}_2\text{O}_6$. Two magnetic transitions at T_{N1} and T_{N2} are marked in the inset.

in Fig. 2(a). An obvious broad maximum at ~ 38 K resembles the behavior of a linear-chain Heisenberg antiferromagnet, in agreement with the chainlike structural features [32]. In fact, similar low-dimensional features have also been observed in other pyroxenes, such as $\text{NaCrGe}_2\text{O}_6$ [33]. With further decreasing temperature, a drop around 11.6 K occurs. As marked by the dashed and dash-dotted lines in the inset, two distinct magnetic transitions at $T_{N1} = 11.6$ K and $T_{N2} = 13$ K can be clearly seen in Fisher's heat capacity $d(\chi T)/dT$, suggesting two magnetically ordered states. This result is consistent with previous studies [14,19].

Experimental magnetic susceptibility was fitted with the Curie-Weiss law between 200 and 350 K. This yields an effective moment $\mu_{\text{eff}} = 6.16(8)\mu_B$, consistent with the calculated spin-only value of $5.92\mu_B$ for the Fe^{3+} cations in the high-spin state, in agreement with the previous report [14]. The negative Weiss temperature of $\Theta = -117(1)$ K indicates predominant antiferromagnetic interactions and reveals a considerable reduction in the Néel temperature, $\Theta/T_N \simeq 10$, which may be due to the low-dimensionality and/or frustration.

In order to further characterize these magnetic phase transitions, we measured the heat capacity of $\text{NaFeGe}_2\text{O}_6$, shown in Fig. 2(b). The two successive cusps at 11.6 and 13 K are indicative of two magnetic phase transitions, in good agreement with our magnetic susceptibility data. No apparent anomaly can be observed around 38 K, implying that the

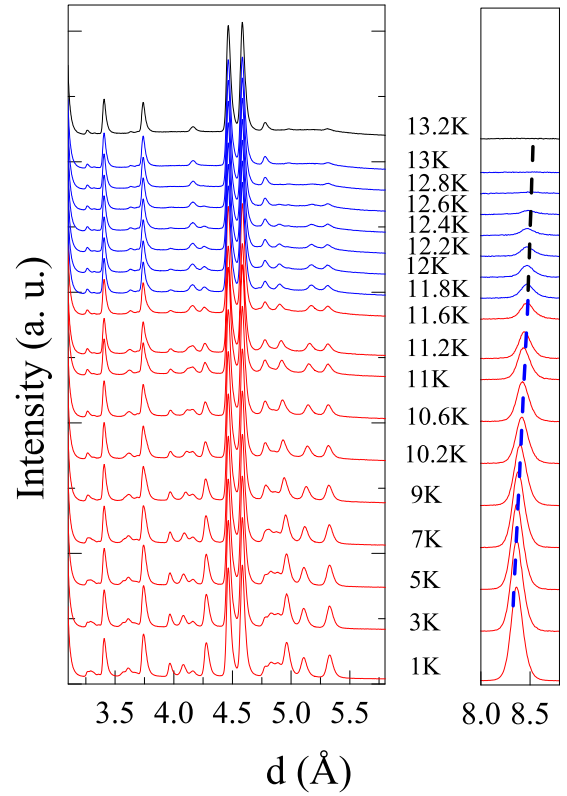


FIG. 3. Temperature dependence of the neutron powder diffraction data of $\text{NaFeGe}_2\text{O}_6$. Black, blue, and red reflections correspond to the paramagnetic, ICM2, and ICM1 phases, respectively.

broad maximum at 38 K should be attributed to short-range magnetic correlations.

B. Neutron diffraction

According to our temperature-dependent x-ray diffraction and the WISH backscattering data collected in the temperature range of 1.5–150 K, $\text{NaFeGe}_2\text{O}_6$ crystallizes with the $C2/c1'$ symmetry and has no symmetry change down to 1.5 K. At 150 K, the lattice parameters are $a = 10.0092(1)$ Å, $b = 8.9124(1)$ Å, $c = 5.50895(5)$ Å, $\beta = 107.5189(9)^\circ$. Magnetic Bragg reflections appear below $T_{N2} = 13$ K (ICM2 phase) in the NPD data, as shown in Fig. 3, and they can be indexed by an incommensurate propagation vector $\mathbf{k} \simeq (\alpha, 0, \gamma)$, with $\alpha = -0.6999(8)$ and $\gamma = 0.0649(2)$ at 12.2 K. The value of \mathbf{k} shows a slightly temperature dependent behavior, as indicated in Fig. 4. On further cooling, additional magnetic reflections appear below $T_{N1} = 11.6$ K (ICM1 phase), and the magnetic reflections exhibit an obvious temperature-dependent behavior. These reflections can also be indexed by the same incommensurate vector \mathbf{k} , albeit with slightly different α and γ values [Figs. 4(a) and 4(b)]. At 1.5 K, the refined \mathbf{k} is $(-0.6702(1), 0, 0.08028(5))$. It is clear that the ICM2 phase appears only within the very narrow temperature range 11.6–13 K.

Previous neutron diffraction experiments on both powder and single crystals failed to resolve this phase [18]. The presence of magnetic Bragg reflections in our neutron diffraction data is consistent with the magnetic susceptibility

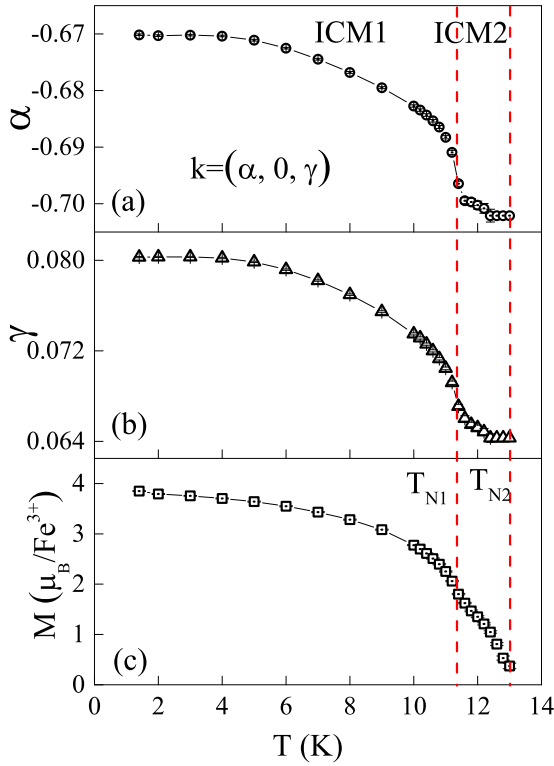


FIG. 4. Temperature dependence of the (a) α and (b) γ components of the magnetic propagation vector and (c) the refined magnetic moment.

and heat capacity measurements, showing the existence of two ordered magnetic states. The propagation vector of the ICM1 phase we obtained is, in fact, equivalent to the vector $\mathbf{k}' = (0.323, 1.0, 0.08)$ reported in Ref. [18]. By applying a reciprocal translation $(-1, -1, 0)$, one can transform \mathbf{k}' into $\mathbf{k} = (-0.67, 0, 0.08)$.

Symmetry analysis was performed in order to determine the magnetic structures of NaFeGe₂O₆. Starting with the parent space group $C2/c1'$ and propagation vector $\mathbf{k} \simeq (\alpha, 0, \gamma)$ in the B plane of the Brillouin zone, two active magnetic irreducible representations, mB1 and mB2, and their corresponding subgroups were obtained using ISODISTORT. For the ICM1 phase, we found that the magnetic superspace group $Cc1'(\alpha, 0, \gamma)0s$ [basis = $(-1, 0, 0, 0), (0, -1, 0, 0), (0, 0, -1, 0), (0, 0, 0, 1)$, origin = $(0, 0, 0, 0)$], generated from the single active mB1 irreducible representation, can be adopted to describe the magnetic structure. Such a symmetry fixes the phase difference between atoms Fe1 $(0, y, 0.25)$ and Fe2 $(0, -y, 0.75)$ at $(1 + \gamma)\pi$. The magnetic structure refinement at 1.5 K was carried out by taking into account this symmetry constraint. The final refinement is shown in Fig. 5, arriving at the cycloidal configuration with magnetic moments in the (ac) plane. The refined total magnetic moment at 1.5 K is $3.857(8)\mu_B$, considerably smaller than the $5\mu_B$ expected for $S = \frac{5}{2}$ of Fe^{3+} . In fact, this value is very close to the total magnetic moment with $4.09(4)\mu_B$ refined from the single-crystal experiment of Ref. [18]. Such a reduction, observed very often in cycloidal spin systems, is likely a consequence

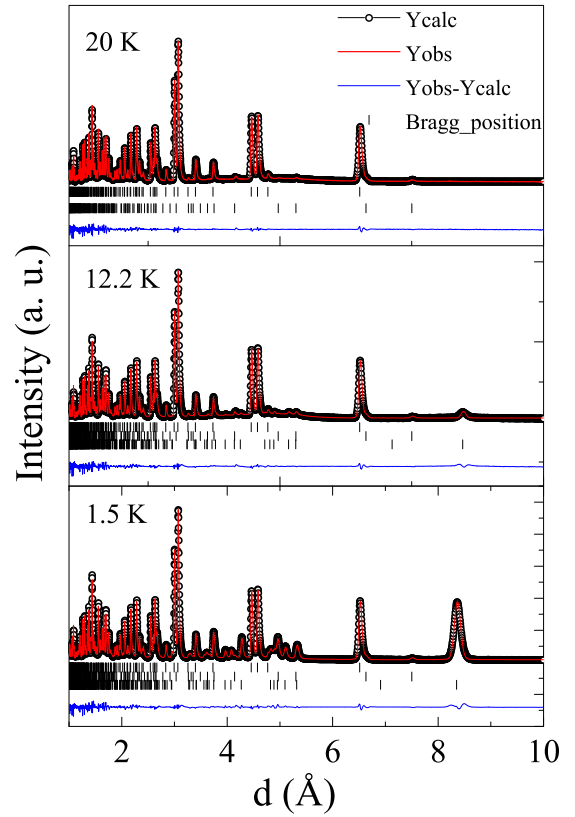


FIG. 5. Neutron powder diffraction patterns for NaFeGe₂O₆ at 20, 12.2, and 1.5 K. The nuclear reflections are denoted by the upper tick marks. The reflections marked in the second line belong to an impurity phase Na₄Ge₉O₂₀ with a weight fraction of 1.52(2)%. The lowest tick marks show magnetic phase (in the case of 12.2 and 1.5 K).

of spin fluctuations and covalency. The magnetic symmetry, as represented in Fig. 1, breaks the inversion symmetry and preserves the mirror-plane symmetry perpendicular to the unique b axis, leading to the magnetic point group $m1'$, which allows the existence of a ferroelectric polarization. Indeed, this magnetic symmetry corroborates the earlier observations of multiferroicity [14, 15].

The magnetic symmetry for the ICM2 phase belongs to the same irreducible representation, but with a distinct magnetic order parameter direction $(a, 0)$. This corresponds to the magnetic superspace group $C2/c1'(\alpha, 0, \gamma)00s$ [basis = $(-1, 0, 0, 0), (0, -1, 0, 0), (0, 0, -1, 0), (0, 0, 0, 1)$, origin = $(0, 0, 0, 0)$], which conserves the inversion symmetry and the twofold screw axis. We found that a sinusoidally modulated magnetic structure is suitable to refine our neutron data at 12.2 K. The refinement leads to a spin moment of $1.55(2)\mu_B$ along the a axis. The final refined neutron diffraction pattern is shown in Fig. 5, and the corresponding magnetic configuration is illustrated in Fig. 1. One can immediately see that it does not break the space inversion and gives rise to a centrosymmetric magnetic point group $2/m1'$. Such a magnetic structure cannot lead to any long-range electric polarization, in agreement with the previous polarization measurements. The temperature-dependent ordered moment of NaFeGe₂O₆ is shown in Fig. 4(c), where the magnetic

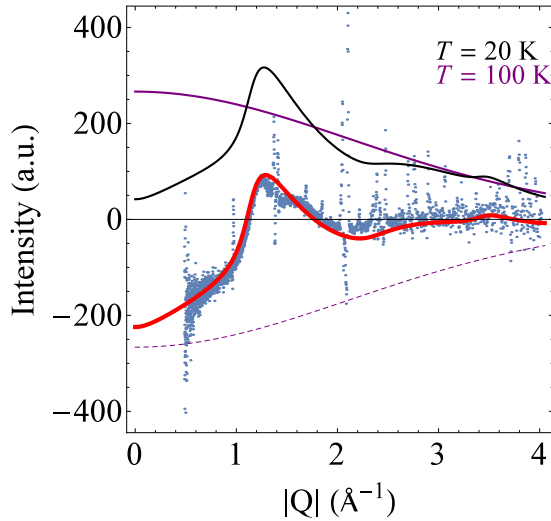


FIG. 6. The difference between neutron diffraction patterns at 20 and 100 K and the fitted curve based on the 1D model [Eq. (2)].

moment for the ICM2 phase [spin-density wave (SDW)] is taken as a quadratic mean of the refined moment.

Having resolved the long-range magnetic order in NaFeGe₂O₆, we now look into the short-range order above T_{N2} . As shown in Fig. 6, the magnetic diffuse scattering in NaFeGe₂O₆ extracted from the difference of the neutron diffraction data collected at 20 and 100 K shows a maximum around $d = 5.1 \text{ \AA}$ (1.4 \AA^{-1}), signaling the presence of short-range magnetic correlations. The feature that sharply rises at low Q and gradually decreases toward high Q is characteristic of one-dimensional spin-spin correlations expected within the structural chains of NaFeGe₂O₆. In the family of magnetic pyroxenes, the presence of one-dimensional correlations has been evidenced in CaMnGe₂O₆ through the analysis of neutron diffuse scattering data based on an analytical one-dimensional antiferromagnetic (AFM) model [17],

$$S(Q) = f(Q)^2 \sum_i \langle S_0 S_i \rangle \frac{\sin(QR_i)}{QR_i}, \quad (2)$$

where $f(Q)$ is the magnetic form factor of Fe³⁺ in the dipole approximation and R_i represents the distance between the sites along the chain. The exponential decrease and the AFM spin-spin correlations $\langle S_0 S_i \rangle$ with the distance d_i and correlation length ξ are expressed as

$$\langle S_0 S_i \rangle = (-1)^i S^2 \exp\left(-\frac{d_i}{\xi}\right). \quad (3)$$

We fitted such a model against the experimental data, with the best fit shown in Fig. 6. The correlation length of $8.0 \pm 0.4 \text{ \AA}$ indicates the short-range magnetic correlations along the c axis. In fact, the onset temperature of the one-dimensional (1D) spin-spin correlations is likely higher than 38 K (the position of the magnetic susceptibility maximum), as weak diffuse scattering is still present at 50 K. Additional evidence for the 1D spin correlations above T_{N2} is obtained from thermal expansion. As shown in Fig. 7, the temperature-dependent lattice volume of NaFeGe₂O₆ refined from the XRD data exhibits apparent negative thermal expansion below

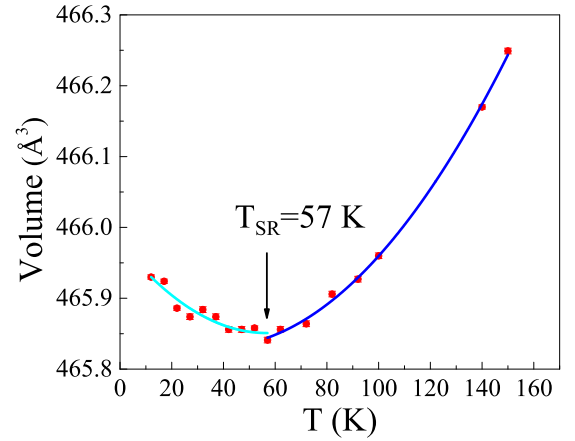


FIG. 7. Temperature dependence of the lattice volume of NaFeGe₂O₆ from variable-temperature XRD.

57 K. This anomaly can be attributed to the magnetostriction effect related to the short-range magnetic order in one dimension [17].

C. Mean-field analysis

As we confirm with the direct *ab initio* analysis in Sec. III D, the crystal structure of NaFeGe₂O₆ hosts three exchange couplings. J_1 runs along the chains of the FeO₆ octahedra, whereas J_2 and J_3 couple these chains into the three-dimensional (3D) network. In the following, we use the mean-field and classical spin approximation that proved efficient in previous studies [34–36] and investigate ordered spin configurations arising from the interplay of J_1 , J_2 , and J_3 .

Consider the primitive cell and the spin Hamiltonian given by

$$H = \sum_{i,j} \sum_{R,R'} J_{ij}^{RR'} S_i^R S_j^{R'}, \quad (4)$$

where $J_{ij}^{RR'}$ is the exchange interaction between the spins S_i and S_j . We employ the method of Freiser [34] to determine the ground state. Suppose σ_i^R represents the mean spin at site i in a cell with the lattice vector R . Then the ordered spin configuration can be expressed in terms of the Bloch spin functions

$$\sigma_i^R = \sum_k \sigma_i^k e^{-ikR}, \quad (5)$$

and the spin-spin interaction energy ξ_{ij} between the two sites becomes

$$\xi_{ij} = \sum_R J_{ij}^R e^{-ikR}. \quad (6)$$

The diagonalization of the quadratic part of the mean-field energy results in the eigenvalue problem

$$\sum_j \left\{ \sum_{R'} J_{ij}^{R'} e^{-ikR'} \right\} \sigma_j = \lambda(k) \sigma_i. \quad (7)$$

The eigenvalues are inversely proportional to the possible transition temperatures, whereas the corresponding eigenvectors yield periodicities of the spin configurations. For

TABLE I. Contributions to the exchange matrix from the spin exchange paths between atom sites Fe1 (0, 0.9036(2), 0.25) and Fe2 (0, -0.0964(2), 0.75) in a primitive setting.

S_i	S_j	$d(\text{Fe-Fe})$ (Å)	\mathbf{R}	Contribution to ξ_{ij}
Fe1	Fe1	6.69	(1,0,0)	$J_3 e^{-ik_x}$
			(-1, 0, 0)	$J_3 e^{ik_x}$
			(0,1,0)	$J_3 e^{-ik_y}$
			(0, -1, 0)	$J_3 e^{ik_y}$
Fe1	Fe2	3.257	(0,0,0)(0,0,1)	$J_1(1 + e^{-ik_z})$
			(-1, 0, 0)(0, -1, 1)	$J_2(e^{ik_x} + e^{i(k_y - k_z)})$
Fe2	Fe1	3.257	(0, 0, 0)(0, 0, -1)	$J_1(1 + e^{ik_z})$
			(1, 0, 0)(0, 1, -1)	$J_2(e^{-ik_x} + e^{i(-k_y + k_z)})$
Fe2	Fe2	6.69	(1,0,0)	$J_3 e^{-ik_x}$
			(-1, 0, 0)	$J_3 e^{ik_x}$
			(0,1,0)	$J_3 e^{-ik_y}$
			(0, -1, 0)	$J_3 e^{ik_y}$

a given set of exchange parameters, one finds the vector \mathbf{k} that delivers the lowest eigenvalue of the interaction matrix. This eigenvector will usually indicate the periodicity of the first (lowest-temperature) ordered state [36,37].

The spin-spin exchange energies associated with the matrix ξ_{ij} from various spin exchange paths J_1 , J_2 , and J_3 in NaFeGe₂O₆ are summarized in Table I. We employed the program ENERMAG [38] to diagonalize the exchange matrix and considered only the AFM case because the couplings J_2 and J_3 are long range and are unlikely to be ferromagnetic, whereas J_1 is known to be AFM too [13,39]. The magnetic ground state depends on relative values of the exchange parameters, so we set $J_1 = 1$ and analyze the magnetic structure as a function of J_2/J_1 and J_3/J_1 (Fig. 8). Each of the interchain couplings taken alone yields commensurate order but of different types, $\mathbf{k} = 0$ in the case of J_2 and $\mathbf{k} = (0, 1, 0)$ in the case of J_3 . The incommensurate phase appears when both J_2 and J_3 are sizable as a result of the competition between the interchain couplings.

The $k = 0$ phase is common to transition-metal pyroxenes and has been reported, e.g., for NaCrGe₂O₆, NaCrSi₂O₆ [33,40], and CaMnGe₂O₆ [17]. It corresponds to

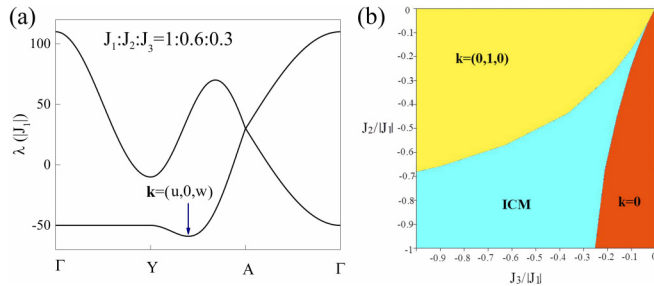


FIG. 8. (a) Dispersion relations for the eigenvalues of the exchange matrix with $J_1 : J_2 : J_3 = 1 : 0.6 : 0.3$ along some lines of symmetry in the first Brillouin zone of the $C2/c1'$ space group. (b) Magnetic phase diagram representing the stability of different magnetic ground states at various exchange parameters. The yellow region shows the magnetic phase characterized by $\mathbf{k} = (0, 1, 0)$, while the region in red corresponds to the $\mathbf{k} = 0$ phase. The phase denoted with cyan denotes an ICM phase.

TABLE II. Isotropic exchange couplings J_i (in K) in NaFeGe₂O₆ as obtained from DFT+ U calculations with different values of the on-site Coulomb repulsion parameter U_d . The last row is the Curie-Weiss temperature Θ (in K).

	$d_{\text{Fe-Fe}}$	$U_d = 6 \text{ eV}$	$U_d = 7 \text{ eV}$	$U_d = 8 \text{ eV}$
J_1	3.25	15.0	12.3	10.2
J_2	5.64	4.5	3.8	3.1
J_3	6.70	2.5	2.1	1.9
Θ		-144	-119	-100

the ferromagnetic ordering of antiferromagnetic spin chains. The $\mathbf{k} = (0, 1, 0)$ state was reported for CaM(Si, Ge)₂O₆ ($M = \text{Fe, Co, Ni}$), where spins are ferromagnetically coupled within chains and antiferromagnetically aligned between the chains. As for NaFeGe₂O₆, its incommensurate order is naturally ascribed to the competition between J_2 and J_3 . Using the $J_1 : J_2 : J_3 = 1 : 0.6 : 0.3$ regime, we find $\mathbf{k} = (-0.6, 0, 0.19)$, in reasonable agreement with the experimental propagation vector from NPD. Note, however, that at this point we analyze only the periodicity of the magnetic structure and cannot distinguish between, e.g., the cycloid and spin-density wave.

D. Microscopic analysis

For a more quantitative and material-specific description of the magnetic ordering, we proceed to the *ab initio* evaluation of the exchange couplings. Several sets of crystallographic data were reported for NaFeGe₂O₆ [18]. We performed DFT calculations for all of them and found only minor differences in the exchange parameters. The effect of the Hubbard U_d is more pronounced, but it pertains to absolute values of J and does not change their hierarchy (Table II).

By evaluating the exchange couplings in the crystallographic unit cell of NaFeGe₂O₆ (four magnetic atoms) and in the doubled cell (eight magnetic atoms), we established that the three exchanges, J_1 – J_3 considered above, are sufficient for the minimum microscopic descriptions, as further long-distance interactions are well below 0.1 K. The resulting couplings are summarized in Table II and can be juxtaposed with the experiment by calculating the Curie-Weiss temperature,

$$\Theta = -\frac{S(S+1)}{3} \sum_i z_i J_i = -\frac{35}{6} (J_1 + J_2 + 2J_3), \quad (8)$$

where z_i stands for the number of couplings per Fe site. The Θ values in Table II show the best agreement with the experiment for $U_d = 7 \text{ eV}$, which yields $J_2/J_1 = 0.31$ and $J_3/J_1 = 0.17$. On the structural level, this hierarchy follows the increase in the Fe-Fe distances. We also note that J_2 involves the double GeO₄ bridge (two tetrahedra linking the FeO₆ octahedra), whereas in the case of J_3 only a single bridge is involved. For comparison, in Cr-based pyroxenes the interactions via the double tetrahedral bridges are predominant as well [39].

Given the two couplings J_1 , two couplings J_2 , and four couplings J_3 per Fe site, NaFeGe₂O₆ should be far from magnetic one-dimensionality because $J_{\text{inter}}/J_{\text{intra}} = (J_2 + 2J_3)/J_1 = 0.66$. On the other hand, J_2 and J_3 form triangular

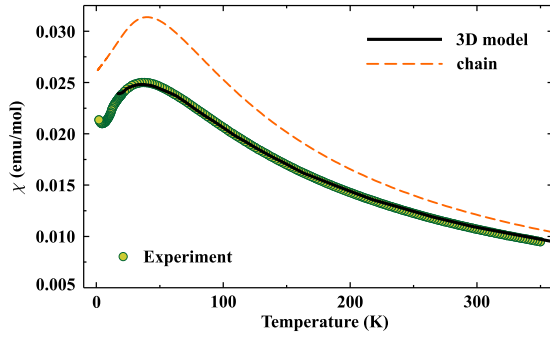


FIG. 9. Fit of the magnetic susceptibility with the 3D spin model including J_1 , J_2 , and J_3 , as explained in the text. The susceptibility of a spin chain with $g = 2.0$ and the same J_1 as in the 3D fit is shown with the dashed line for reference.

loops and compete (Fig. 1). This competition can also be seen from the fact that J_2 and J_3 stabilize different types of the interchain order. The fact that $2J_3 = 4.2$ K is similar to $J_2 = 3.8$ K renders $\text{NaFeGe}_2\text{O}_6$ strongly frustrated. This frustration not only triggers the incommensurate ordering but also introduces short-range order in the paramagnetic state, as we present below.

Regarding the long-range ordered state, simple classical minimization leads to an incommensurate state with the propagation vector $\mathbf{k} = (-0.675, 0, 0.09)$, in good agreement with the experimental $\mathbf{k} = (-0.67, 0, 0.08)$ at 1.5 K. Monte Carlo simulations produce magnetic susceptibility with a broad maximum, and the magnetic transition takes place well below this maximum, a signature of short-range correlations above T_N . By keeping the J_2/J_1 and J_3/J_1 ratios from DFT and adjusting J_1 as well as other parameters, we arrive at the best fit with $g = 1.99$ and $J_1 = 9.6$ K, which corresponds to the susceptibility maximum at 38 K and $T_N \simeq 12$ K. Note that this model features only one magnetic transition because no anisotropy terms are involved.

We also calculated magnetic susceptibility for a single spin chain with the same value of $J_1 = 9.6$ K and $g = 2.0$. As shown in Fig. 9, it reproduces the overall shape of the experimental susceptibility data, but the absolute values do not match. This confirms that the susceptibility maximum in $\text{NaFeGe}_2\text{O}_6$ is related to the magnetic one-dimensionality, yet the interchain couplings are clearly non-negligible.

Finally, we estimate the single-ion magnetic anisotropy. To this end, we fix spins along a given direction and rotate the reference spin in the plane perpendicular to this direction [30]. This yields a as the magnetic easy axis. Placing the reference spin along b and c increases the energy by 0.50 and 0.62 K, respectively, leading to an effective single-ion anisotropy of $A \simeq 0.09$ K and $z = a$ in Eq. (1). This weak anisotropy is similar in size to that of other Fe^{3+} oxide compounds [29]. The easy-axis anisotropy naturally explains the formation of the SDW state with spins along a in the ICM2 phase because in the presence of anisotropy a collinear structure is preferred at elevated temperatures over a noncollinear one [41]. The cycloid in the ICM1 phase features a component along the a direction too, which is consistent with the calculated single-ion anisotropy.

IV. DISCUSSION

Transition-metal pyroxenes show variable magnetic dimensionality and different types of the long-range order. The dimensionality changes between quasi-1D and 3D depending on the tetrahedral group [39], whereas several flavors of commensurate and collinear long-range order were reported in previous studies [17,33,40]. Some of the pyroxenes show signatures of the frustration, such as the enhanced ratio Θ/T_N between the Curie-Weiss and Néel temperatures, but this reduction in T_N is typically related to the magnetic one-dimensionality [39]. $\text{NaFeGe}_2\text{O}_6$ reveals a distinct microscopic scenario, where frustration is present and plays a central role. The competing interactions J_2 and J_3 are well balanced and trigger incommensurate magnetic order, which is uncommon to pyroxenes. Despite the sizable interchain interactions, $\text{NaFeGe}_2\text{O}_6$ shows signatures of 1D magnetism above T_{N2} because the chains are effectively decoupled. We note in passing that a similar microscopic scenario may be relevant to $\text{SrMnGe}_2\text{O}_6$, where an incommensurate magnetic structure was revealed by neutron diffraction [16].

Another distinct feature of $\text{NaFeGe}_2\text{O}_6$ is its two consecutive magnetic transitions at T_{N1} and T_{N2} . The majority of pyroxenes show only one magnetic transition, as expected in nonfrustrated antiferromagnets. The frustration itself, the competition between J_2 and J_3 , does not split the transition into two, and the presence of weak single-ion anisotropy seems to be crucial here. Similar combinations of the cycloid and SDW phases were observed in systems like $\text{Ca}_3\text{Co}_2\text{O}_6$ [42,43] and $\text{Li}_2\text{NiW}_2\text{O}_8$ [44], where magnetic ions bear strong single-ion anisotropy. Although Fe^{3+} with its half-filled d shell is by far less anisotropic than Ni^{2+} or Co^{3+} , the anisotropy energy of less than 1 K (and less than 1% of the leading exchange coupling J_1) is already sufficient for driving similar physics. The main difference is the incommensurate and noncollinear ground-state magnetic configuration stabilized by the isotropic exchange couplings J_i in $\text{NaFeGe}_2\text{O}_6$, whereas in systems with stronger anisotropy, commensurate and collinear states favored by the anisotropy occur.

Altogether, we resolved the earlier controversies regarding the magnetic structures of $\text{NaFeGe}_2\text{O}_6$, established the microscopic magnetic model of this compound, and outlined the microscopic condition for the formation of incommensurate spin states in transition-metal pyroxenes (Fig. 8). Let us now discuss the multiferroic behavior of $\text{NaFeGe}_2\text{O}_6$ from the symmetry perspective of the magnetic structures determined in this work.

The magnetic superspace group $Cc1'(\alpha, 0, \gamma)0s$ of the ICM1 phase breaks both spatial inversion and time reversal. This cycloidal magnetic symmetry allows electric polarization within the (ac) plane [Fig. 10(b)], in good agreement with the experimental observation. The polarization can be explained by the theory of the inverse DM effect or spin-current model. However, this mechanism does not account for the observation of a small polarization (less than $2 \mu\text{C}/\text{m}^2$) along the b axis in a synthetic single crystal [15]. In principle, the symmetry analysis of $\text{NaFeGe}_2\text{O}_6$ allows the presence of a magnetic moment along the b axis and indicates that both cycloidal and proper-screw components might be present, as illustrated in Fig. 10(a). We also examined other recently

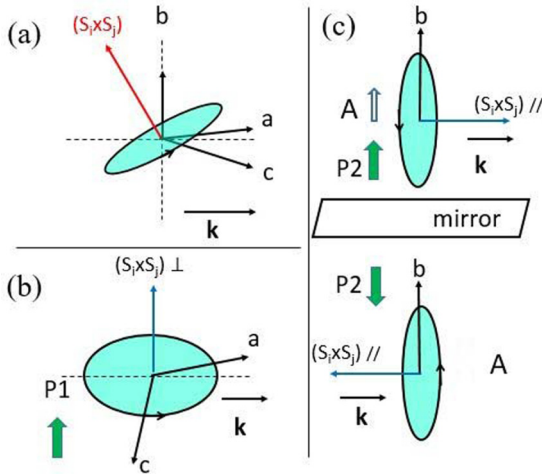


FIG. 10. (a) The relationship between the spin helicity vector and the propagation vector \mathbf{k} assuming the presence of the b -spin component. (b) The projection of the spin helicity vector onto the direction perpendicular to the vector \mathbf{k} . (c) The projection of the spin helicity vector onto the direction parallel to the vector \mathbf{k} and its mirror-plane-related one.

developed mechanisms for explaining multiferroicity in materials showing proper-screw magnetic symmetry. The cycloidal spin arrangement [Fig. 10(b)] based on $\mathbf{P} \propto (\mathbf{A} \cdot \mathbf{r}_{ij})(\mathbf{S}_i \times \mathbf{S}_j)$ gives no electric polarization along the b axis because the mirror plane contains \mathbf{r}_{ij} [6,10,45,46]. In light of the ferroaxial mechanism, the proper-screw component can lead to $\mathbf{P}_2 \propto \mathbf{A} \cdot \{\mathbf{r}_{ij} \cdot (\mathbf{S}_i \times \mathbf{S}_j)\}$ along the b axis. However, the mirror plane perpendicular to the b axis leads to the opposite spin chirality, as explained schematically in Fig. 10(c), leading to the cancellation of the electric polarization. As a result, the magnetic superspace group requires that the electric polarization can be present only in the (ac) plane. A signal in the pyrocurrent measurement along the b direction could then be due to a misalignment of the crystal or an impurity phase, such as hematite (α -Fe₂O₃) and maghemite (γ -Fe₂O₃), which were identified in the crystal on which the pyrocurrent measurement of Ref. [15] was performed.

V. CONCLUSION

In conclusion, we presented the revisited magnetic structures and associated microscopic magnetic model for NaFeGe₂O₆. This compound shows a cycloid magnetic configuration below 11.6 K preceded by a spin-density-wave state below 13 K and a region of one-dimensional spin-spin correlations extending up to at least 50 K. Competing interchain couplings J_2 and J_3 decouple the spin chains above T_N and render magnetic order incommensurate below T_N . The cycloid phase is a direct result of this competition, whereas the SDW phase should form upon the presence of weak single-ion anisotropy that tends to align the spins along the a direction. We reported the general magnetic phase diagram of transition-metal pyroxenes, which captures all the documented magnetic structures reported in pyroxenes so far, and argued that the electric polarization of NaFeGe₂O₆ should be confined to the ac plane within the cycloid phase, whereas no electric polarization should occur within the SDW phase, which is centrosymmetric.

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